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# Magnetic relaxation and lower critical field in MgB<sub>2</sub> wires

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## Abstract

Magnetic relaxation behaviour, critical current density  $J_c$  and lower critical field  $H_{c1}$  have been investigated in MgB<sub>2</sub>/Ta/Cu wires. It is found that  $J_c$  and  $H_{c1}$  decrease linearly with temperature in the whole temperature region below  $T_c$ . The relaxation rate is very small and has a weak temperature dependence compared to high- $T_c$  superconductors. Also, the pinning potential is much larger and the temperature and field dependences of the pinning potential are briefly discussed.

## 1. Introduction

The discovery of the new superconductor MgB<sub>2</sub>, with a critical temperature  $T_c$  close to 40 K, has generated a lot of interest in the field of superconductivity due to its high  $T_c$  and good superconducting properties [1]. The transition temperature of MgB<sub>2</sub> is much higher than Nb<sub>3</sub>Ge (by almost a factor of two), having the highest  $T_c$  in conventional superconductors. Some work has been done recently, including the observation of an isotope effect and band structure studies, which suggest that MgB<sub>2</sub> is a conventional phonon-mediated BCS superconductor [2, 3]. The advantage of MgB<sub>2</sub> is its applications in a higher-temperature region (20–30 K) due to its high  $T_c$ , low cost and absence of a weak link, where conventional superconductors cannot play a role due to their low  $T_c$ s. Also, a high upper critical field of 29–39 T has been reported in this material [4]. Recently, a very high  $J_c$  of  $1.2 \times 10^7$  A cm<sup>-2</sup> at 4.2 K at zero field was obtained in the *in situ* epitaxial MgB<sub>2</sub> thin film, suggesting that MgB<sub>2</sub> can reach extremely high intrinsic  $J_c$  [5]. These results indicate that MgB<sub>2</sub> may be the most promising compound for large-scale applications. Therefore, many groups are fabricating

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MgB<sub>2</sub> wires and tapes with high performance by using different metal sheaths. Canfield *et al* [6] fabricated high-density MgB<sub>2</sub> wires (160 μm in diameter) through the exposure of boron filaments to Mg vapour. Also, the recent reports on the preparation of MgB<sub>2</sub> wires by powder-in-tube (PIT) using either Ag or Cu sheaths and MgB<sub>2</sub> strands by filling Nb-lined monel tubes with commercial MgB<sub>2</sub> powders were the first steps to putting MgB<sub>2</sub> superconductors into applications [7, 8]. By using Cu as a sheath, a transport  $J_c$  of 50 000 A cm<sup>-2</sup> at 15 K in a self-field was obtained in MgB<sub>2</sub> wires. For the non-sintered MgB<sub>2</sub>/Ni tape,  $J_c$  reached around 10<sup>5</sup> A cm<sup>-2</sup> at 4.2 K in a self-field [8]. Recently, Wang *et al* [9] reported their results on an Fe-clad MgB<sub>2</sub> wire, in which  $J_c$  achieved  $4.2 \times 10^5$  A cm<sup>-2</sup> at 4.2 K in a self-field.  $J_c$  is further improved to  $1.7 \times 10^4$  A cm<sup>-2</sup> in 1 T at 29.5 K and at 33 K in a self-field [10]. Also, transport  $J_c$  of 8700 and 55 830 A cm<sup>-2</sup> at 4.2 K in a self-field were measured for Cu- and Fe/Cu-sheathed MgB<sub>2</sub> square wires, respectively, by using commercial MgB<sub>2</sub> powder [11]. On the other hand, another important feature that should be considered is the magnetic relaxation behaviour of MgB<sub>2</sub> superconductors. It is well known that a rapid decay of critical currents was often observed in high temperature superconductors (HTSC), which hampers its applications. To date, there are almost no reports on the magnetic relaxation behaviour of MgB<sub>2</sub> wires and tapes. In this paper, we present experimental results on the temperature dependence of  $J_c$ , magnetic relaxation behaviour and the field dependence of the lower critical field  $H_{c1}$  in MgB<sub>2</sub>/Ta/Cu wires. Our data indicate that  $J_c$  and  $H_{c1}$  decrease linearly with temperature and that the relaxation rate is very small compared to high- $T_c$  superconductors.

## 2. Experimental details

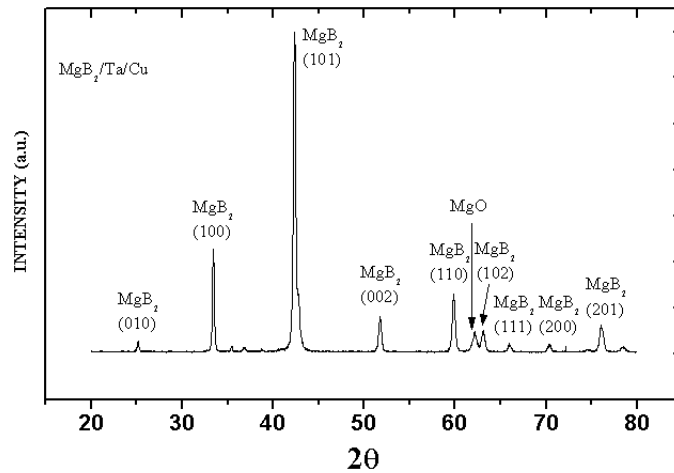
Single filamentary MgB<sub>2</sub>/Cu composite wires with Ta as a buffer layer were fabricated using the *in situ* powder-in-tube process. Mg powder and amorphous B powder were used as starting materials in an atomic ratio of Mg:B = 1:2. A proportion of 5% extra Mg was added to compensate the loss of magnesium at high temperature. A detailed description of this process can be found elsewhere [12]. Finally, the wires were sintered at 600–900 °C for 2 h in argon at ambient pressure.

The dc magnetization was performed using a SQUID magnetometer. All the magnetization measurements were measured by first cooling the sample in zero field and then applying a field to begin the measurement. The field was applied perpendicular to the axis of the sample. After it finished at a given temperature, the field was set to zero and the temperature was warmed to 50 K to completely remove the trapped field inside the sample. The phase composition was analysed by x-ray diffraction measurements. A critical temperature  $T_c$  of 38.4 K with a sharp transition width of 0.6 K was observed from the magnetization measurements in the sample.

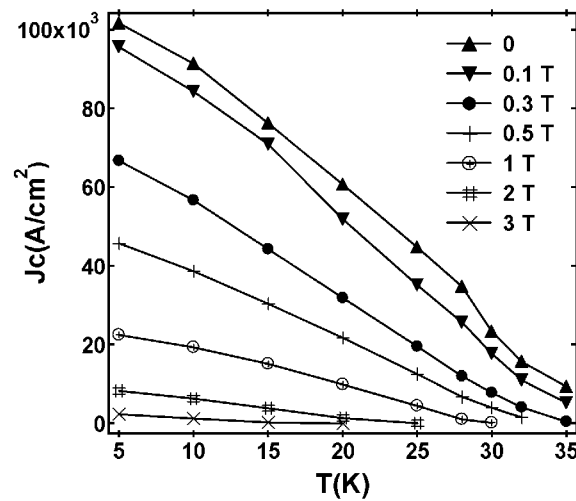
## 3. Results and discussion

X-ray diffraction patterns for the MgB<sub>2</sub>/Ta/Cu wires are given in figure 1. The line-widths of the peaks are sharp, indicating that the sample exhibits good crystallinity. As seen in figure 1, the MgB<sub>2</sub> grains are not well textured. Also, no MgB<sub>4</sub> phase is found in the sample, which has a negative effect on the superconducting properties of MgB<sub>2</sub> due to its large mismatch in the crystal structure. In addition, only a few MgO phases can be observed in the sample.

Figure 2 shows the temperature dependence of the magnetic  $J_c$  at various fields for MgB<sub>2</sub>/Ta/Cu wire. The  $J_c$  values are calculated from the magnetization curves using Bean's model of  $J_c = 30\Delta M/d$ , where  $d$  is the diameter of the sample and  $\Delta M$  is the width of the magnetic hysteresis loops. It can be observed that  $J_c$  is relatively high around 10<sup>5</sup> A cm<sup>-2</sup> at 5 K in a self-field and  $J_c$  decreases quickly as temperature increases. Interestingly, the



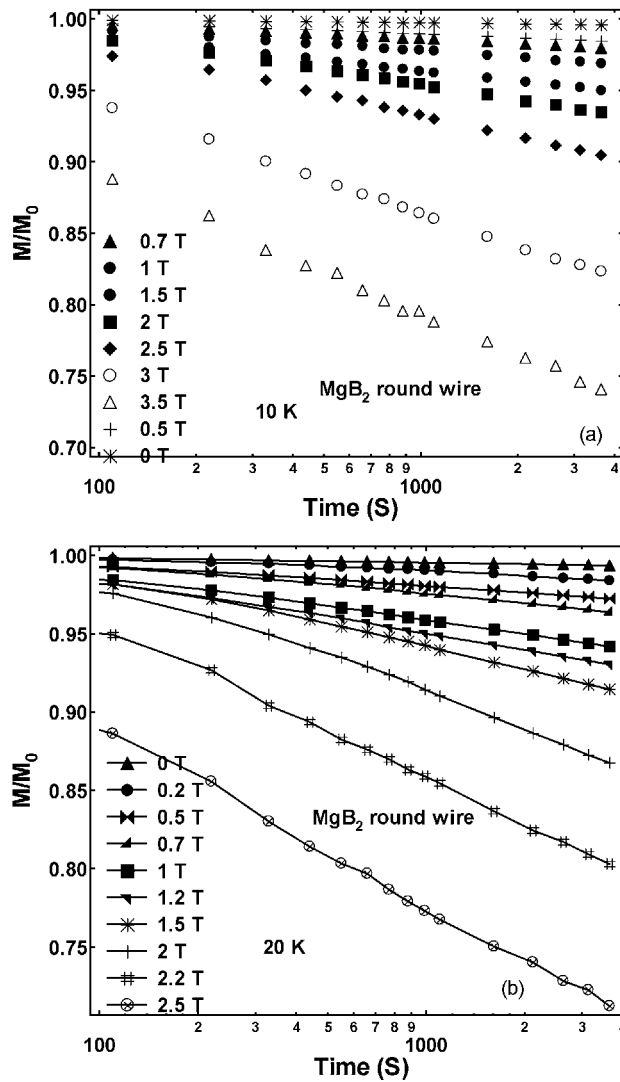
**Figure 1.** An x-ray diffraction pattern of the MgB<sub>2</sub>/Ta/Cu wire. No MgB<sub>4</sub> is observed and only a few MgO phases exists.



**Figure 2.** The critical current density as a function of temperature in various fields from 0.1 to 3 T. It is observed that  $J_c$  decreases with temperature almost linearly.

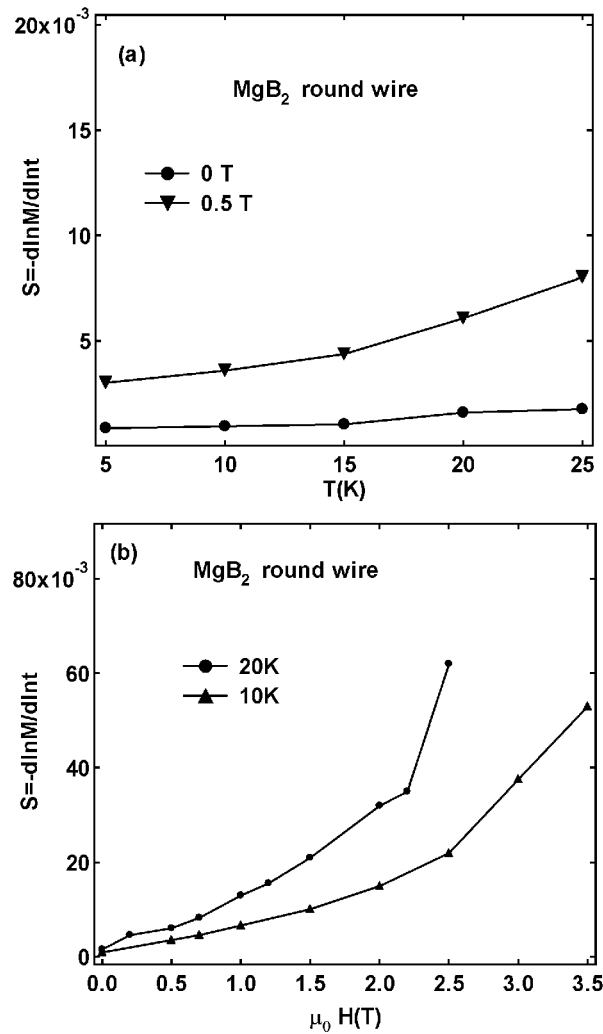
sample exhibits a simple temperature dependence, where  $J_c$  drops linearly with temperature at various fields. Similar behaviour was also found in the MgB<sub>2</sub> bulk samples [13]. In high- $T_c$  superconductors,  $J_c$  often decreases with temperature according to a quasi-exponential law, which may be related to the large thermally activated flux creep. Therefore, it can be assumed that the thermal flux creep is not serious in MgB<sub>2</sub> superconductors.

The decay of normalized magnetization with time  $t$  was measured for the MgB<sub>2</sub>/Ta/Cu wires using the SQUID magnetometer, and figure 3 shows the results. It can be observed that the normalized magnetization exhibits the linear decay of  $\ln t$  at various temperatures and fields, indicating that this behaviour follows the Anderson–Kim model. The decay rate  $S$  was calculated from the equation  $S = -d \ln M / d \ln t$ . Figure 4 presents the decay rate as a function of field at 20 and 10 K and as a function of temperature at 0 and 0.5 T. The relaxation rate is very



**Figure 3.** The time dependence of the normalized magnetization at (a) 10 K and (b) 20 K for  $MgB_2/Ta/Cu$  wires. A good linear relationship between the normalized magnetization and  $\ln t$  is found, suggesting that the Anderson–Kim model is effective.

small, just around 0.086% at 5 K in 0 T. When the field increases to 0.5 T,  $S$  just reaches 0.003, being the same as the  $MgB_2$  bulk sample [13, 14]. This value is an order of magnitude lower than that of HTSC. Furthermore, the relaxation rate shows a very weak temperature dependence in the measured temperature range. At 0.5 T,  $S$  increases from 0.003 at 5 K to 0.008 at 25 K. As described by Thompson *et al* [13], the exponent  $n$  of the electric field  $E = E_0(J/J_0)^n$  is simply estimated by  $(n - 1) = 1/S$ , where  $J$  is the current density. This will lead to an  $n$  value of around 300, corresponding to very steep curves in transport  $I$ – $V$  characteristic. On the other hand, the relaxation rate increases slowly with field and may be high at a higher field, as observed in figure 4. For example, at 20 K,  $S$  increases from 0.0046 at 0.2 T to 0.062 at 2.5 T. The very small relaxation rate and weak temperature dependence may be related to the strong



**Figure 4.** The relaxation rate as a function of temperature at (a) 0 and 0.5 T and (b) as a function of field at 10 and 20 K.

pinning barrier. The pinning potential  $U_0$  of the sample can be calculated from the expression based on the Anderson–Kim model of  $U_0 = -kT/S$ . In figure 5, we present  $U_0$  as a function of field and temperature.  $U_0$  does not change with temperature monotonically, while it decays monotonically with field. It can be observed that the pinning potential at 0 T is much higher than that at 0.5 T at various temperatures. However,  $U_0$  is almost the same at 10 and 20 K in high fields above 1 T. On the other hand, the pinning potential of the new superconductor is very large compared to HTSC. It is concluded that the intrinsic pinning energy is very high and that the pinning well of MgB<sub>2</sub> superconductors is so deep that the thermal activation and fluctuation has a trivial effect [15, 16].

Figure 6 shows the field dependence of magnetization at various temperatures for MgB<sub>2</sub>/Ta/Cu wires. It is evident that all the curves exhibit the common linear field dependence of magnetization induced by the Meissner effect at low fields. The linear field dependence of magnetization is observed in fields below 400 Oe at 5 K. The lower critical field  $H_{c1}$  is

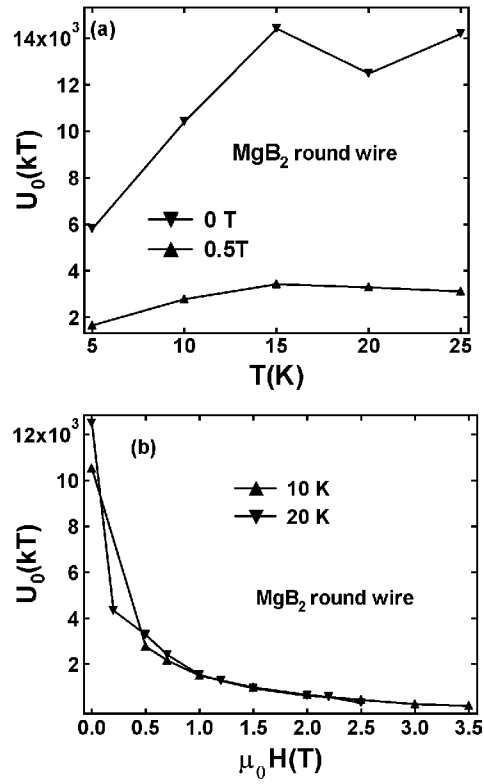


Figure 5. The pinning potential  $U_0$  versus temperature at (a) 0 and 0.5 T and (b) versus field at 10 and 20 K.

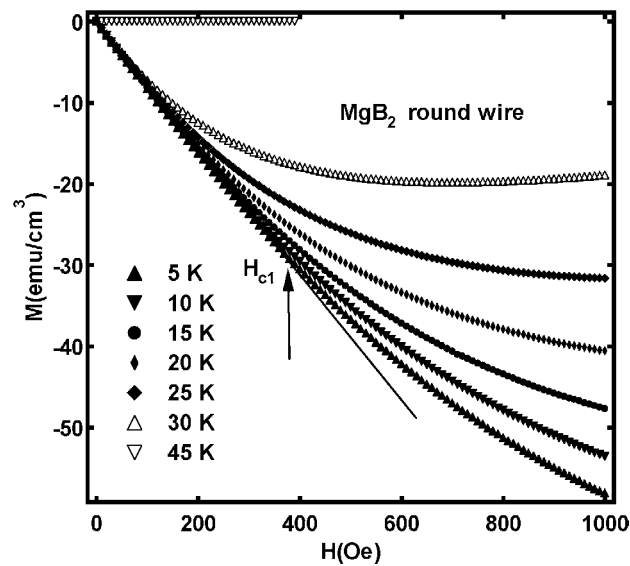
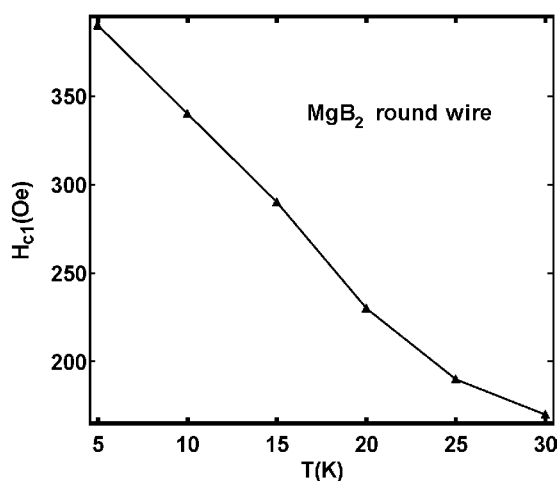


Figure 6. The field dependence of magnetization at various temperatures for  $\text{MgB}_2/\text{Ta}/\text{Cu}$  wire. It is found that the initial slope of all the curves is the same.



**Figure 7.** The lower critical field as a function of temperature, showing a linear temperature dependence.

determined by the departure point from the linearity on the slope of the magnetization curve and the results are displayed in figure 7. The criterion of 10% of deviation from linearity was used for the determination of  $H_{c1}$ . At 5 K,  $H_{c1}$  is around 390 Oe, which is higher than that of MgB<sub>2</sub> bulk samples [17]. Also, the lower critical field shows the linear dependence on temperature, which is similar to that in the YNi<sub>2</sub>B<sub>2</sub>C system and in high- $T_c$  superconductors. It is believed that this behaviour may be related to the linear dependence of the upper critical field in MgB<sub>2</sub>, as reported by some authors [18]. The linear temperature dependence of  $H_{c1}$  is considered to be in contrast to an isotropic s-wave superconductivity in MgB<sub>2</sub> [17].

#### 4. Conclusion

In conclusion, we have investigated the phase composition, temperature dependence of  $J_c$ , magnetic relaxation behaviour and lower critical field in MgB<sub>2</sub>/Ta/Cu wires. The sample has a critical temperature of 38.4 K and a  $J_c$  value of  $10^5$  A cm<sup>-2</sup> at 5 K in a self-field. In contrast to high- $T_c$  superconductors,  $J_c$  is found to decrease linearly with temperature. Magnetic relaxation measurements indicate that the relaxation rate is very small, with a weak temperature dependence, resulting in a high pinning potential. Also, a linear temperature dependence of the lower critical field has been observed in the whole temperature range. Our results provide some useful information for the applications of MgB<sub>2</sub> superconductors.

#### Acknowledgments

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